

# The physical properties of SnO<sub>2</sub> thin films deposited by rheotaxial growth and thermal oxidation method

Shatha Shammon Batros Jami

Ministry of Science and Technology, Baghdad, Iraq.

Accepted 21 November, 2014

---

## ABSTRACT

This research SnO<sub>2</sub> thin films was prepared by using rheotaxial growth and thermal oxidation (RGTO) method. The pure Sn powder was evaporate by using melbedume bot at vacuum 10<sup>-6</sup> torr on glass substrate till it reached its evaporate temperature. Thin films at thickness 3000<sup>o</sup>A were prepared, after that treated with 450°C and to obtain SnO<sub>2</sub> thin films in the presence of O<sub>2</sub>. The X-ray Diffraction (XRD) results refer to polycrystalline phase Scanning Electron Microscopy (SEM), which studies the local morphology and the surface of SnO<sub>2</sub> thin films obtained by Rheotaxial Growth and thermal Oxidation (RGTO) method. SEM shows a fractal-like morphology of nanograins (22 nm typical size). The optical properties were studying by using uv-vis spectra and it was found that the transmittance was ~70%. The value of energy gap was 3.85 eV.

**Keywords:** RGTO thin films, thin dioxide thin film.

E-mail: shathajammel@yahoo.com.

---

## INTRODUCTION

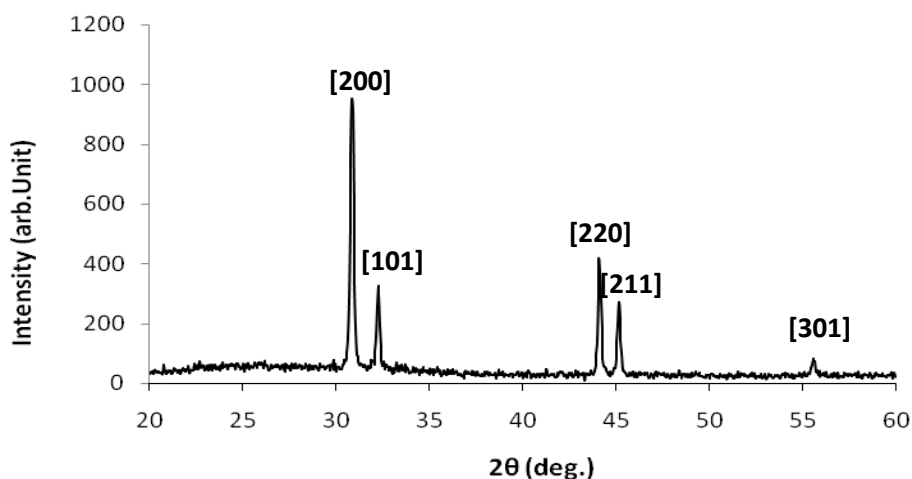
Tin dioxide (SnO<sub>2</sub>) serves as an important base material in a variety of resistive-type gas sensors (Katsiev et al., 2008). The widespread applicability of this semiconducting oxide is related both to its range of conductance variability and to the fact that it responds to both oxidizing and reducing gases (Barsan et al., 1999). Till now, commercial gas sensors devices (mainly used for domestic applications in the gas alarm systems) based on SnO<sub>2</sub> have been fabricated with thick (about 1 mm) films, of which a fundamental limitation is large power consumption (Ihokura and Watson, 1994). This limitation does not concern the thin solid film gas sensors (Batzill et al., 2003; Batzill et al., 2004; Batzill et al., 2006; Batzill and Diebold, 2006; Katsiev et al., 2007).

Among various techniques for the preparation of SnO<sub>2</sub> thin films the Rheotaxial Growth and Thermal Oxidation (RGTO) method first proposed by Sberveglieri (Batzill et al., 2004; Batzill et al., 2006) looks like one of the most effective methods in view of gas sensing applications. This technique is based on a two-step process. It consists of: i) deposition of a thin Sn film on a substrate

maintained at a temperature slightly exceeding the Sn melting point (231.8°C) (first step); and ii) a subsequent long-term post-deposition oxidation at high temperature up to 700°C (second step). In the first step, a discontinuous Sn film is formed composed of isolated, almost spherical droplets. Due to the absence of percolation in the plane, electrical resistivity of the thin film is infinite. The granular structure is preserved during the final oxidation but the Sn droplets undergo a significant increase in size due to oxygen incorporation. This results in an almost 30% increase of the film thickness. Moreover, the second step, via oxidation, creates connecting "necks" between the formerly isolated Sn droplets, and such necks open a percolation path for the flow of electric current in the RGTO SnO<sub>2</sub> based gas sensing devices (Sberveglieri et al., 1990; Sberveglieri, 1992; Sberveglieri et al., 1992). In present work, we propose a microscopic investigation of such optimized RGTO samples. X-ray Diffraction (XRD) data of SnO<sub>2</sub> sample is presented and discussed in comparison with Scanning Electron Microscopy (SEM) images and the

**Table 1.** Summary of deposition conditions.

Coating unit	INFICON V90
Materials	Tin powder 99.99%
Substrates	glass slides
Vacuum	$\sim 6 \times 10^{-6}$ torr
Substrate to film gap	18 cm

**Figure 1.** X- ray diffraction of Sn metallic.

optical properties of  $\text{SnO}_2$  prepared in RGTO are studied.

## EXPERIMENTAL

Tin dioxide ( $\text{SnO}_2$ ) thin films were deposited according to the RGTO method. In the first step, thin metallic tin layers were deposited by classical vacuum thermal evaporation. The evaporation was carried out in a conventional vacuum coating unit (INFICON V90) under a vacuum of order of  $6 \times 10^{-6}$  torr with controlled deposition rate by using pure Sn. A summary of the deposition conditions is shown in Table 1. The second step, involving the thermal oxidation of Sn thin layer, was performed for 4 h at  $450^\circ\text{C}$  in dry air atmosphere inside a reaction chamber of a diffusion furnace. The crystallinity and the crystalline phases present in the films were checked by XRD using an X-ray diffractometer with  $\text{CuK}\alpha$  radiation (model "Shimadzu XRD 6000"). For the different conditions of preparing  $\text{SnO}_2$  sample, recorded was for a range of  $2\theta$  from  $10^\circ$  to  $60^\circ$  at  $2^\circ$  glancing angle.

The local surface morphology were investigated by scanning electron microscopy (SEM), model (VEGA3 TESCAN). The absorbance and transmittance of the deposited thin film was measured using UV-VIS spectrum (Optima Sp – 300 Plus) in the wavelength region of 200 to 1100 nm.

## RESULTS AND DISCUSSION

The RGTO thin film crystalline structure has been investigated with XRD. Figure 1 shows a typical Bragg-Brentano spectrum taken on an identically prepared film grown on glass. Diffraction data show the presence of the [110], [101], [200], [211] and [220] diffraction peaks from crystalline  $\text{SnO}_2$  in the cassiterite rutile phase. The diffraction pattern perfectly matches with the  $\text{SnO}_2$  reference crystal structure 1997 JCPD file v.1:30. Figure 2 is the main diffraction line from a coexisting SnO phase which should be observed when the heat treatment of Sn is about  $250^\circ\text{C}$  as shown in Figures 1 and 2. Thus our results are in line with what already reported in literature (that is, absence of coexisting SnO phase when the heat treatment of Sn is about  $450^\circ\text{C}$  for 4 h) and with the expectation of a complete transition to a  $\text{SnO}_2$  phase upon annealing in oxygen at temperatures above  $450^\circ\text{C}$  as shown in Figure 3 (Dieguez et al., 2000).

The SEM investigation of the RGTO  $\text{SnO}_2$  thin film without annealing shows surface morphology of fractal like morphology of nanograins (22 nm typical size) with homogenous distributions. The crystallography of fractal looks like a rectangular prism with square base and this is one of the classes of Tetragonal Bravais lattices of  $\text{SnO}_2$  as shown in Figure 4.

The optical properties were studied by measuring the

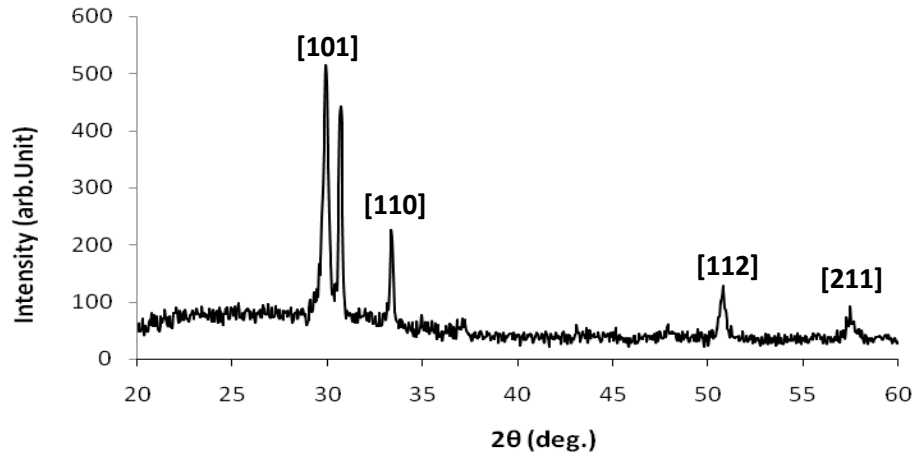


Figure 2. X-ray diffraction of SnO thin films at 250°C oxidation of Sn.

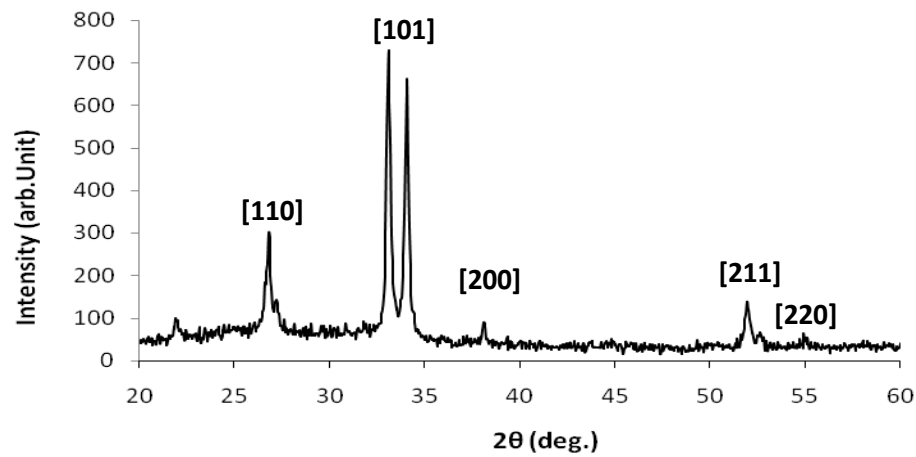
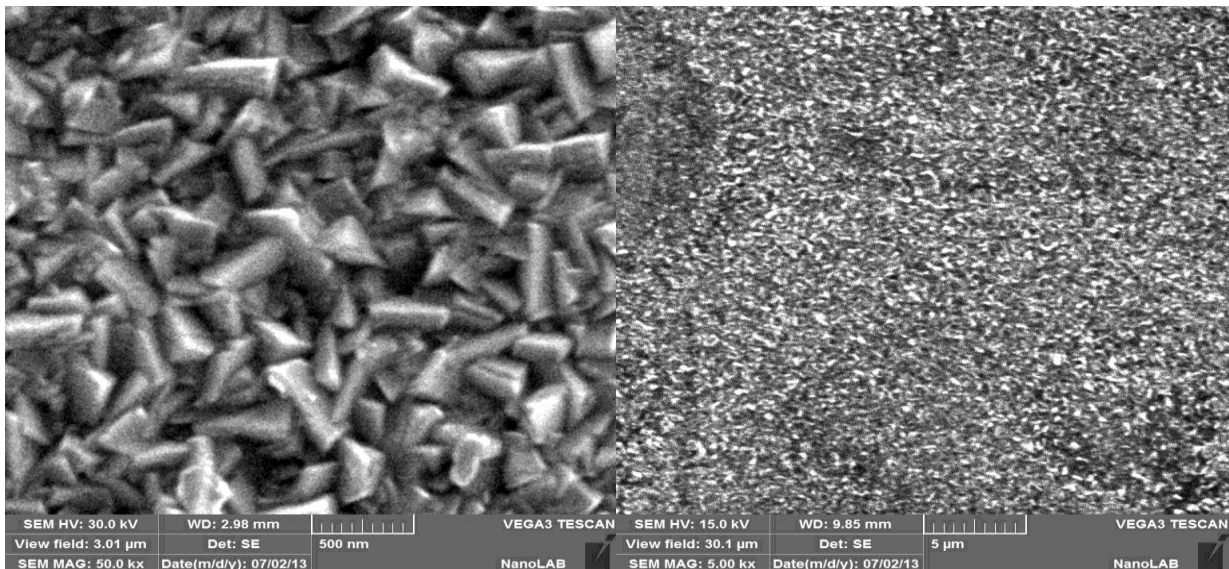
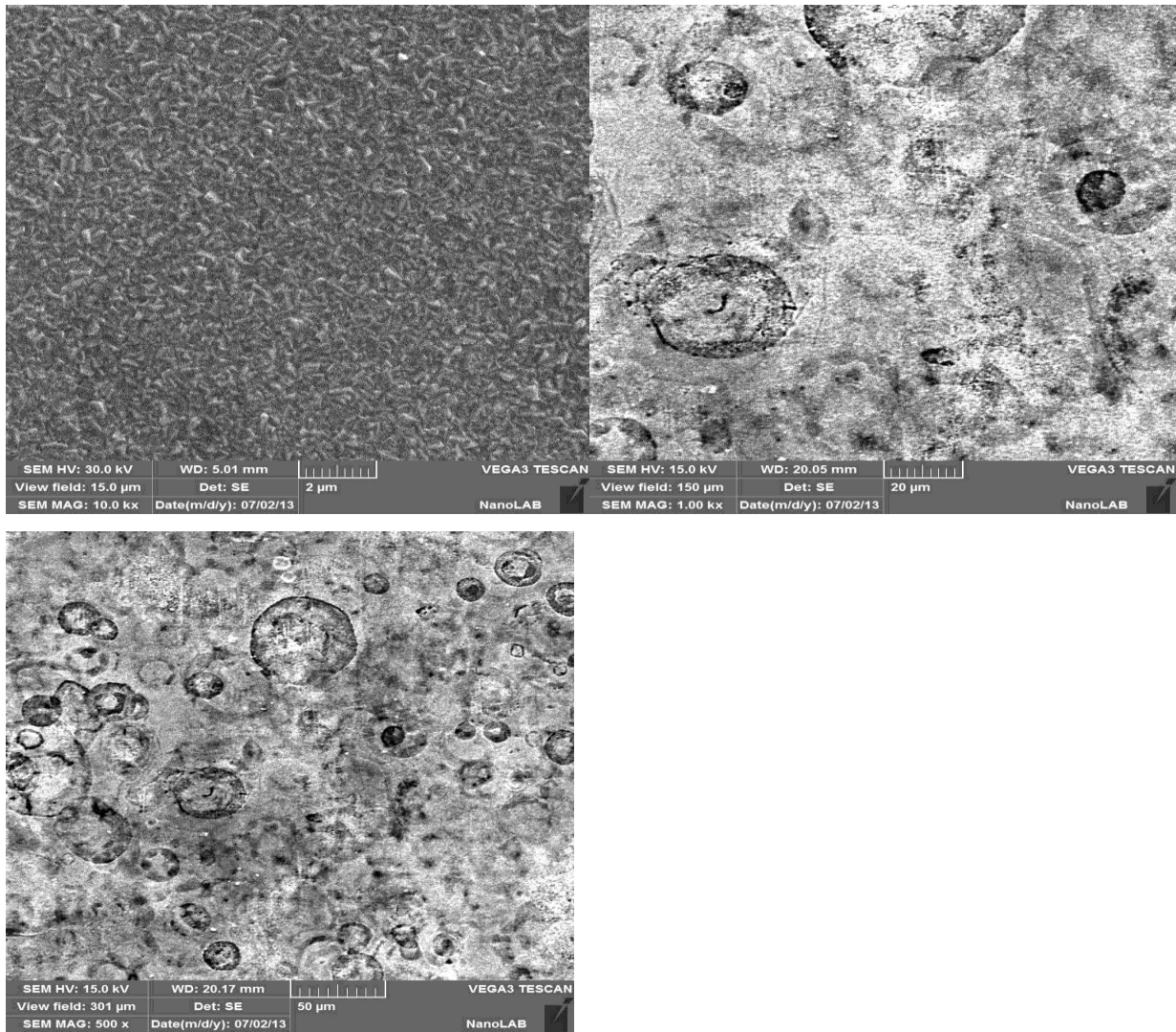
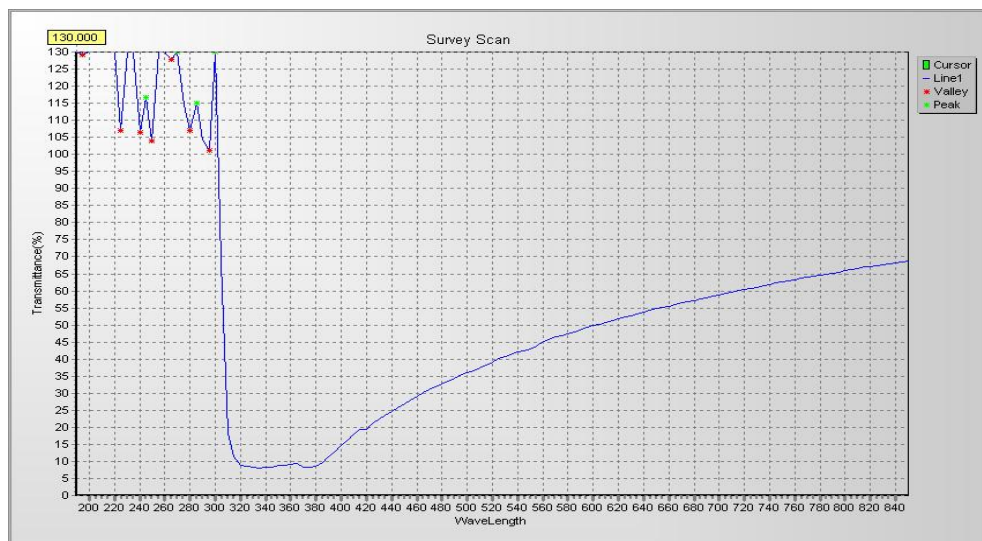


Figure 3. X-ray diffraction of SnO<sub>2</sub> thin films at 450°C oxidation of Sn.





**Figure 4.** From the top to bottom and left to right SEM images at increasing magnification (500kx, 1.00kx, 5.00kx, 10kx, 50kx) of nanograins of SnO<sub>2</sub>.



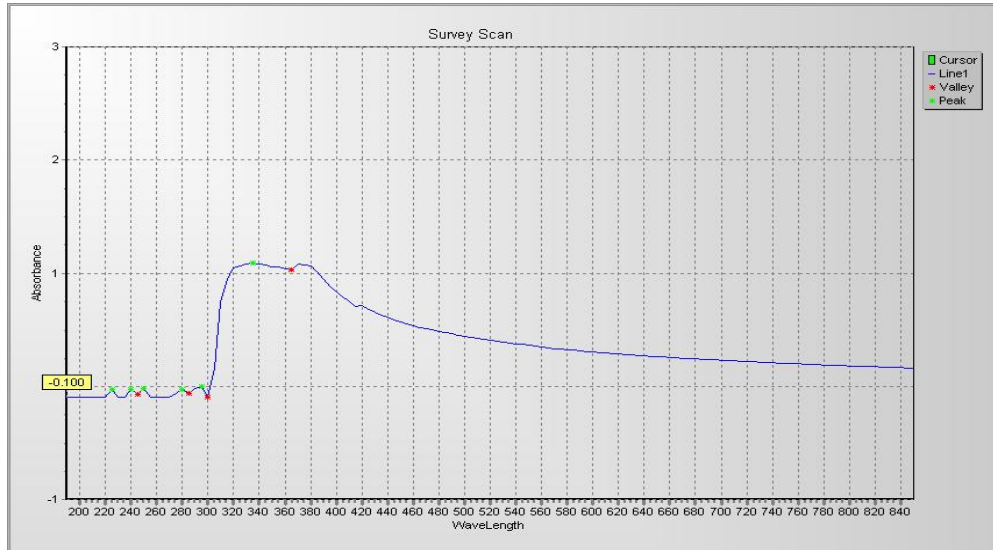


Figure 5. Transmittance and absorbance spectra of SnO<sub>2</sub> films.

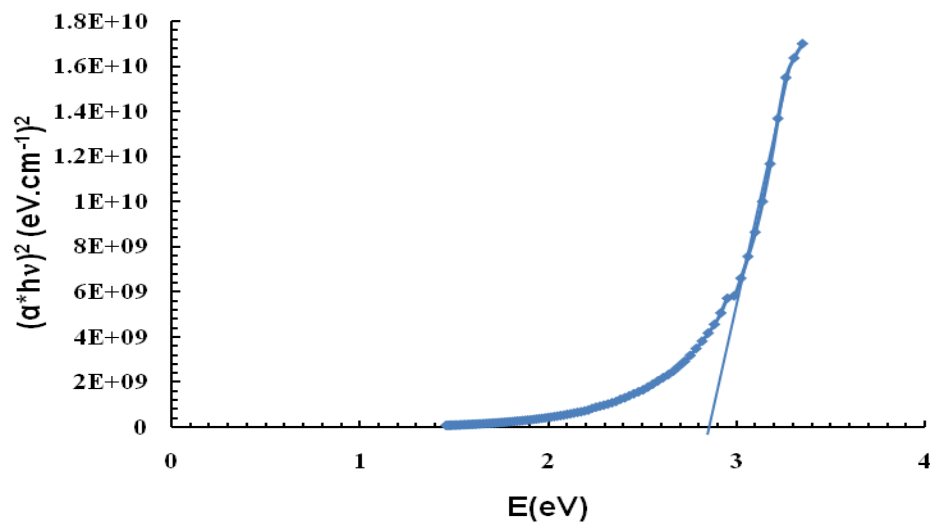


Figure 6. Plot of  $(\alpha hv)^2$  versus  $(hv)$  curve of ZnO:In<sub>2</sub>O<sub>3</sub> thin film.

transmission and absorbance spectra as shown in Figure 5. We found that the film has high transmission at long wave lengths approximately (65 to 70%), and decreasing transmission to 10% at short wavelengths.

The energy gap ( $E_g$ ) was determined by employing the following relation (Kim et al., 2008):

$$\alpha = A(hv - E_g)^n / hv \quad (1)$$

Where  $\alpha$  is absorption coefficient,  $A$  a constant (independent from  $\nu$ ) and  $n$  the exponent that depends upon the quantum selection rules for the particular material. The photon energy ( $h\nu$ ) for y-axis can be calculated using Equation 2.

$$E = h\nu = hc/\lambda \quad (2)$$

Where  $h$  is Plank's constant ( $6.626 \times 10^{-34}$ ),  $c$  is speed of light ( $3 \times 10^8$ ) and  $\lambda$  is the wavelength. The values of energy gap  $E_g$  is 2.85eV as shown in Figure 6.

## CONCLUSIONS

In this paper, we present the results of comparative study of surface morphology of RGTO SnO<sub>2</sub> thin films using the XRD, SEM. XRD study of the atomic structure confirms the existence of SnO<sub>2</sub> polycrystalline phase, and it is found that in 450°C for 4 h the existence of SnO<sub>2</sub> are

without any coexisting SnO phase. SEM study of local surface morphology show a fractal like morphology of nanograins (22 nm typical size). The transmittance of the thin films is ~70%; the value of energy gap was 3.85eV.

## REFERENCES

- Barsan N**, Schweitzer-Barberich M, Göpel W, **1999**. Fundamental and practical aspects in the design of nanoscaled SnO<sub>2</sub> gas sensors: a status report. *Fresenius J Anal Chem*, 365:287-304.
- Batzill M**, Bergermayer W, Tanaka I, Diebold U, **2006**. Tuning the chemical response of a gas sensitive material: Water adsorption on SnO<sub>2</sub>(101). *Surf Sci*, 600:L29.
- Batzill M**, **Diebold U**, **2006**. Characterizing solid state gas responses using surface charging in photoemission: water adsorption on SnO<sub>2</sub>(101). *J Phys: Condensed Matt*, 18:L129-L134
- Batzill M**, Katsiev K, Diebold U, **2003**. Surface morphologies of SnO<sub>2</sub>(110). *Surf Sci*, 529:295-311.
- Batzill M**, Katsiev K, Diebold U, **2004**. Tuning the oxide/organic interface: Benzene on SnO<sub>2</sub>(101). *Appl Phys Lett*, 85:5766-5768.
- Dieguez A**, Romano-Rodriguez A, Morante JR, Sangaletti L, Depero LE, Comini E, Faglia G, Sberveglieri G, **2000**. Influence of the completion of oxidation on the long-term response of RGTO SnO<sub>2</sub> gas sensors. *Sens Actuators B*, 66:40-42.
- Ihokura K**, **Watson J**, **1994**. *The Stannic Oxide Gas Sensor: Principles and Applications*. CRC Press.
- Katsiev K**, Batzill M, Boatner LA, Diebold U, **2008**. Defects and Pd growth on the reduced SnO<sub>2</sub> (100) surface. *Surf Sci*, 602:1699-1704.
- Katsiev K**, Batzill M, Diebold U, Urban A, Meyer B, **2007**. Growth of one-dimensional Pd nanowires on the terraces of a reduced SnO<sub>2</sub>(101) surface. *Phys Rev Lett*, 98:86102.
- Kim H**, Auyeung RY, Piqué A, **2008**. Transparent conducting F-doped SnO<sub>2</sub> thin films grown by pulsed laser deposition. *Thin Solid Films*, 516:5052- 5056.
- Sberveglieri G**, **1992**. Classical and novel techniques for the preparation of SnO<sub>2</sub> thin-film gas sensors. *Sens Actuators B*, 6:239-247.
- Sberveglieri G**, Faglia G, Gropelli S, Nelli P, Camanzi A, **1990**. A new technique for growing large surface area SnO<sub>2</sub> thin film (RGTO technique). *Semicond Sci Technol*, 5:1231.
- Sberveglieri G**, Faglia G, Gropelli S, Nelli P, **1992**. A novel PVD technique for the preparation of SnO<sub>2</sub> thin films as C<sub>2</sub>H<sub>5</sub>OH sensors. *Sens Actuators B*, 7:721-726.

---

**Citation:** Jami SSB, 2014. The physical properties of SnO<sub>2</sub> thin films deposited by rheotaxial growth and thermal oxidation method. *Phys Sci Res Int*, 2(4): 62-67.

---